

- BAUER, S. H. (1967). *Electron Diffraction Studies at High Temperatures*. Report Nonr-401(41), Project NRO92-504, ARPA order No. 23-53.
- BAUER, S. H., KATADA, K. & KIMURA, K. (1968). *Structural Chemistry and Molecular Biology*, Ed. A. RICH and N. DAVIDSON, p. 653. San Francisco: Freeman.
- BAUER, S. H. & KIMURA, K. (1962). *J. Phys. Soc. Japan*, **17**, 300.
- BEWILOGUA, L. (1931). *Phys. Z.* **32**, 740.
- BONHAM, R. A. & BARTELL, L. S. (1959). *J. Chem. Phys.* **31**, 702.
- BONHAM, R. A. & MOMANY, F. A. (1963). *J. Phys. Chem.* **67**, 2474.
- BONHAM, R. A. & UKAJI, T. (1962). *J. Chem. Phys.* **36**, 72.
- COONRADT, H. L., HARTOUGH, H. D. & JOHNSON, G. C. (1948). *J. Amer. Chem. Soc.* **70**, 2564.
- CROMER, D. T., LARSON, A. C. & WABER, J. T. (1963). Los Alamos Scientific Laboratory Reports, LA-2987.
- GRONOWITZ, S. & HOFFMAN, R. A. (1958). *Ark. Kemi*, **13**, 279.
- HEDBERG, K. & IWASAKI, M. (1964). *Acta Cryst.* **17**, 529.
- HENCHER, J. L. & BAUER, S. H. (1967). *J. Amer. Chem. Soc.* **89**, 522.
- KIMURA, K. & KUBO, M. (1960). *J. Chem. Phys.* **32**, 1776.
- LANCASTER, J. E. & STOICHEFF, B. B. (1956). *Canad. J. Phys.* **34**, 1016.
- LANGSETH, A. & STOICHEFF, B. P. (1956). *Canad. J. Phys.* **34**, 350.
- LONGUET-HIGGINS, H. C. (1949). *Trans. Faraday Soc.* **45**, 174.
- MURATA, Y. & MORINO, Y. (1966). *Acta Cryst.* **20**, 605.
- NARDELLI, M., FAVA, G. & ARMELLINI, L. (1958). *Ric. Sci.* **28**, 383; *Structure Reports*, Ed. W. B. PEARSON. Vol. 22, p. 770. Utrecht: Oosthoek.
- PATTERSON, A. L. (1944). *Phys. Rev.* **65**, 195.
- PENFOLD, B. R. (1953). *Acta Cryst.* **6**, 591, 707.
- SCHOMAKER, V. & PAULING, L. (1939). *J. Amer. Chem. Soc.* **69**, 1769.
- STRAND, T. G. (1966). *J. Chem. Phys.* **44**, 1661.

Acta Cryst. (1970). B26, 1020

CuWO₄, a Distorted Wolframite-Type Structure

BY LARS KIHNBORG AND ELIZABETH GEBERT*

Institute of Inorganic and Physical Chemistry, University of Stockholm, S-104 05 Stockholm, Sweden

(Received 17 July 1969)

The crystal structure of CuWO₄, previously reported by the authors to be of a distorted wolframite type, has been refined from single-crystal X-ray diffractometer data. The least-squares procedure including anisotropic thermal parameters resulted in a conventional $R=0.029$. The structure is triclinic, space group $P\bar{1}$, with unit-cell dimensions $a=4.7026 (\pm 6)$, $b=5.8389 (\pm 7)$, $c=4.8784 (\pm 6)$ Å, $\alpha=91.677 (\pm 9)$, $\beta=92.469 (\pm 7)$ and $\gamma=82.805 (\pm 10)^\circ$. The copper atom is surrounded by six oxygen atoms, four of these in an approximately square planar configuration and the remaining two at a longer distance completing an elongated octahedron. The tungsten atom is located within a slightly distorted octahedron but is considerably displaced from its centre. Interatomic distances and angles are given. The refinement gave no evidence for oxygen deficiency and new density measurements have given values which, although still somewhat low, are closer to that expected for stoichiometric composition than was the previously reported value.

Introduction

In a previous communication from this Institute we reported a preliminary investigation of the crystal structure of copper tungstate, CuWO₄, which we found to be a distorted version of the wolframite type (Gebert & Kihlberg, 1967). Density measurements led us to believe that our samples were strongly oxygen deficient; hence we preferred to give the formula as CuWO_{4-x}. The study was based on photographically recorded X-ray diffraction intensities obtained from a twinned crystal. The relatively poor quality of the data limited the degree of accuracy that could be achieved. The final R value was 0.20.

When the preliminary investigation had reached a final stage we found an untwinned single crystal of the phase. This paper reports a refinement of the structure based on diffractometer data obtained from this crystal.

Experimental

The crystal was selected from a sample prepared by heating an equimolar mixture of CuO and WO₃ (reagent grade) in an evacuated platinum tube for 4 days at 800°C. The preparation did not appear homogeneous under the microscope; it contained both light yellow, well-developed crystals and amber-coloured pieces with a glassy appearance. Its powder pattern, however, showed only the lines characteristic for CuWO_{4-x} (Gebert & Kihlberg, 1967) and single-crystal photographs revealed that both types of crys-

* On leave from Argonne National Laboratory, Argonne, Illinois, U.S.A.

tal were structurally identical although all species of the clear yellow type that were examined exhibited polysynthetic twinning. The crystals were twinned across (010).

The unit-cell dimensions given in Table 1 were obtained from a powder pattern recorded in a Guinier-Hägg focusing camera using Cu $K\alpha_1$ radiation and KCl as internal standard ($a=6.2919 \text{ \AA}$). The lattice parameters were refined using a least-squares program written by P.-E. Werner at this Institute. Powder patterns were recorded from several samples prepared but no significant variations in the cell dimensions were observed.

Table 1. *Crystallographic data for CuWO₄*

Symmetry: triclinic, space group $P\bar{1}$	
Unit-cell dimensions:	
$a = 4.7026 (\pm 6) \text{ \AA}$	$\alpha = 91.677 (\pm 9)^\circ$
$b = 5.8389 (\pm 7)$	$\beta = 92.469 (\pm 7)$
$c = 4.8784 (\pm 6)$	$\gamma = 82.805 (\pm 10)$
$V = 132.73 \text{ \AA}^3$	
$Z = 2$	
Density: $D_{\text{meas}} = 7.61 - 7.65 \text{ g.cm}^{-3}$	
$D_{\text{calc}} = 7.790 \text{ g.cm}^{-3}$	

A chemical analysis of the phase gave 20.66% (by weight) copper *versus* the theoretical 20.40% copper content in CuWO₄.

The density of the phase was determined previously by pycnometric measurements and was reported to be $D=7.44 (\pm 7) \text{ g.cm}^{-3}$. This value is rather low compared with that calculated for CuWO₄ (Table 1) and was the main reason for our believing that this phase normally exhibits a considerable oxygen deficiency. New measurements, using a slightly more refined pycnometric technique, have been made on another sample which gave values within the range 7.61–7.65 g.cm^{-3} . Although still low compared with the value calculated for CuWO₄, the difference is not necessarily significant since systematic errors of this sign and magnitude are sometimes encountered in density determinations on fine powders. Considering the light colour and transparency of the crystals as well as the result of the structure refinement, reported below, we do not believe that the specimen used in our investigation was significantly oxygen deficient. The stoichiometry of CuWO₄ under various conditions will be the subject of a separate study at this Institute.

The single-crystal studied was amber-coloured, transparent and roughly prismatic with a length (along **a**) of 0.22 mm and a maximum cross section of 0.085 mm. It was mounted on a manual diffractometer (General Electric Single Crystal Orienter) and angle settings were made for the 1111 reciprocal lattice points for which $l \geq 0$ and $\sin^2 \theta \leq 0.30$ ($d > 0.649 \text{ \AA}$). The intensities were recorded according to the θ - 2θ scan technique using Nb-filtered Mo K radiation.

All intensities for which $I/\sigma(I) < 1.25$ (I =net intensity, σ =standard deviation based on counting statistics) were rejected which left a set of 998 values for use in the refinement.

Refinement

The data were corrected for Lorentz and polarization factors and also for the effect of absorption using a local version of the program of Coppens, Leiserowitz & Rabinovich (1965) modified by O. Olofsson and M. Elfström, University of Uppsala. The total linear absorption coefficient was assumed to be $\mu=539.4 \text{ cm}^{-1}$.

Starting from the parameters reported previously (Gebert & Kihlberg, 1967) the initial least-squares cycles were carried out with use of a block-diagonal matrix program (*SFLS*, written by S. Åsbrink, Univ. of Stockholm and C.-I. Brändén, University of Uppsala, modified for IBM 1800 by B. Brandt, University of Stockholm). A full-matrix program (*LALS*, a local version of the program *UCLALS* by Gantzel, Sparks & Trueblood modified by R. Liminga, J.-O. Lundgren and C.-I. Brändén, University of Uppsala) was used in the final stage.

The HFS atomic scattering factors given by Hanson, Herman, Lea & Skillman (1964) were used and the complex anomalous dispersion correction calculated by Cromer (1965) was applied to the copper and tungsten values in the last cycles of refinement. The least-squares programs used minimize the function $\sum w(|F_o| - |F_c|)^2$ and the weights were calculated according to the expression suggested by Cruickshank,

$$w = (A + |F_o| + C|F_o|^2 + D|F_o|^3)^{-1}.$$

By inspection of the weight analysis the following values of the constants in this formula were chosen for the last cycles of refinement, $A=113$, $C=-0.130$ and $D=0.0030$. The weight analysis obtained in the final cycle is given in Table 2.

Table 2. *Weight analysis*

$\Delta = ||F_{\text{obs}}| - |F_{\text{calc}}||$, w =weighting factor, N =number of independent reflexions. The $w\Delta^2$ values have been normalized.

F_{obs}	N	$w\Delta^2$
0.0–18.7	99	1.74
18.7–27.1	100	0.94
27.1–35.4	100	0.61
35.4–42.6	100	0.73
42.6–47.9	100	0.99
47.9–53.6	99	1.02
53.6–61.3	100	0.95
61.3–70.8	100	0.53
70.8–86.8	100	0.64
86.8–151.2	100	1.86
sin θ		
($\times 10^4$)	N	$w\Delta^2$
0–2553	121	1.94
2553–3216	111	1.37
3216–3682	98	0.47
3682–4052	110	0.71
4052–4365	91	0.86
4365–4639	104	0.78
4639–4883	104	0.85
4883–5106	88	0.63
5106–5310	87	1.04
5310–5500	84	1.10

In the course of the refinement it became evident that the strongest reflexions suffered seriously from extinction since $|F_{\text{obs}}|$ was consistently smaller than $|F_{\text{calc}}|$. All data were then corrected for this effect by applying the formula of Zachariasen (Zachariasen, 1963; Zachariasen & Plettinger, 1965) modified as described by Åsbrink & Norrby (1970). The necessary functions were calculated by the absorption correction program (Åsbrink & Werner, 1966) and the empirical constant entering into the calculations was obtained in the usual way from the discrepancies between F_{obs} and F_{calc} (Åsbrink & Werner, 1966).

Refinement of the extinction corrected data with anisotropic thermal parameters for all atoms resulted in a final reliability index $R = \sum ||F_o| - |F_c|| / \sum |F_o| = 0.029$. The final parameters are given in Tables 3 and 4. Observed and calculated structure amplitudes are listed in Table 5.

Table 3. Fractional atomic coordinates for CuWO₄

All atoms in point position 2(i) of space group $P\bar{1}$. E.s.d.'s are given in brackets.

	x	y	z
Cu	0.49533 (16)	0.65976 (13)	0.24481 (15)
W	0.02106 (4)	0.17348 (3)	0.25429 (4)
O(1)	0.2491 (10)	0.3535 (8)	0.4245 (10)
O(2)	0.2145 (10)	0.8812 (7)	0.4309 (9)
O(3)	0.7353 (10)	0.3803 (8)	0.0981 (9)
O(4)	0.7826 (9)	0.9079 (8)	0.0533 (9)

As mentioned above, there had been some doubt concerning the oxygen content of the phase. Although the isotropic B values obtained in the course of the refinement were normal and did not indicate vacancies on any specific oxygen position – the B_{O} values were equal within their standard deviations – a few refinement cycles were performed in which an occupancy parameter for each oxygen position was also refined. In this case the program *ORFLS* by Busing, Martin & Levy (1962) was used. The positional parameters did not change significantly and the following values were obtained for the occupancy parameters (standard deviations in parentheses): 0.94 (± 4), 1.01 (± 4),

0.89 (± 4) and 1.03 (± 4) for the oxygen atom positions 1 through 4. The deviations from unity are hardly significant and cannot be used as a proof for oxygen deficiency. The B_{O} values, which are strongly correlated with the corresponding occupancy parameters (correlation coefficients 0.79–0.82), were much more divergent after this refinement.

Description

As mentioned above, our preliminary investigation of this crystal structure suggested that it is a distorted version of the wolframite type and this is definitely confirmed by the present refinement.

The wolframite type structure is adopted by a number of tungstates of AWO_4 stoichiometry, A being a sufficiently small divalent cation. At least four compounds with this structure have been studied by single-crystal methods, namely NiWO₄ (Keeling, 1957), CdWO₄ (Chichagov, Ilyukhin & Belov, 1966), MnWO₄ (Dachs, Stoll & Weitzel, 1967) and FeWO₄ (Ülkü, 1967). The last structure has been refined from X-ray and neutron diffraction data to a high degree of accuracy and is well suited for a comparison with CuWO₄.

The normal wolframite structures are monoclinic, space group $P2_1/c$, with both metal atoms situated on twofold axes. They can be described as a framework of oxygen atoms in an approximately hexagonal close-packing with the cations occupying half of the octahedral sites. Cations of each kind form layers which are interleaved between the oxygen sheets in alternating sequence (Fig. 2). Infinite zigzag chains are formed by edge-sharing octahedra containing like cations (Fig. 1).

The above description applies also to the CuWO₄ structure. The more irregular coordination required by the copper atoms compared with iron, for example (see below), is achieved primarily by a shear (not to be confused with 'crystallographic shear') parallel to \mathbf{b} along each copper plane by which the oxygen layers surrounding the copper atoms become slightly displaced with respect to each other. This is evident in Figs. 1 and 2. The displacement destroys the twofold symmetry and is directly reflected in the deviation of the angle γ from 90°.

Table 4. Thermal parameters

The β values refer to the temperature factor $\exp [-(h^2\beta_{11} + k^2\beta_{22} + l^2\beta_{33} + 2hk\beta_{12} + 2hl\beta_{13} + 2kl\beta_{23})]$.

$R(i)$, $i=1,3$, are the r.m.s. components of thermal displacement (in Å) along the principal axes of the ellipsoid of thermal vibration (calculated with the program *ORFFE* by Busing, Martin & Levy, 1964). E.s.d.'s in parentheses.

	β_{11} ($\times 10^5$)	β_{22} ($\times 10^5$)	β_{33} ($\times 10^5$)	β_{12} ($\times 10^5$)	β_{13} ($\times 10^5$)	β_{23} ($\times 10^5$)	$R(1)$	$R(2)$	$R(3)$
Cu	587 (27)	429 (18)	428 (27)	184 (17)	-159 (22)	-109 (18)	0.060 (5)	0.066 (4)	0.110 (3)
W	452 (11)	238 (7)	272 (10)	26 (5)	-79 (6)	17 (5)	0.051 (2)	0.063 (1)	0.078 (1)
	($\times 10^4$)	($\times 10^4$)	($\times 10^4$)	($\times 10^4$)	($\times 10^4$)	($\times 10^4$)			
O(1)	85 (16)	55 (10)	61 (14)	-4 (10)	-13 (12)	-12 (10)	0.075 (20)	0.093 (16)	0.110 (17)
O(2)	91 (16)	34 (9)	57 (15)	24 (10)	2 (13)	-8 (10)	0.057 (24)	0.085 (14)	0.116 (14)
O(3)	85 (16)	50 (10)	45 (14)	19 (10)	-4 (12)	-8 (10)	0.070 (19)	0.077 (21)	0.117 (15)
O(4)	79 (15)	60 (10)	35 (13)	3 (10)	0 (12)	2 (10)	0.065 (13)	0.088 (15)	0.107 (14)

are 1.807 and 2.075 Å. There is thus a close similarity between the W–O bond lengths in these structures despite the difference in symmetry.

The copper coordination is subjected to the Jahn–Teller distortion typical for divalent copper and which gives rise to a considerable lengthening of two opposite Cu–O bonds. The details of the coordination should be evident from Table 6 and Fig. 3. The same type of distortion, even more pronounced, is present in CuO. In that case the Cu–O distances are 1.9509 (± 26), 1.9608 (± 13) and 2.7840 (± 37) Å, each occurring twice (Åsbrink & Norrby, 1970). In CuMoO₄, of a structure entirely different from the present one (Abrahams, Bernstein & Jamieson, 1968), two of the three non-equivalent copper atoms have a similar 4+2 coordination, while the third is five-coordinated. The equatorial Cu–O distances around the former atoms range from 1.933 to 1.980 Å, with an average of 1.956 Å which is almost exactly the same as in CuO. The corresponding average in CuWO₄ is 1.976 Å. The long Cu–O bonds in CuMoO₄ fall within the range 2.213–2.619 Å with an average of 2.312 Å, only slightly shorter than in the tungstate. Cu₃WO₆ has five-coordinated copper atoms with the oxygen atoms forming a somewhat distorted trigonal bipyramid (Gebert & Kihlberg, 1969). Here, the five Cu–O distances range from 1.921 to 2.243 Å with a next-nearest oxygen atom not closer than 3.10 Å.

All oxygen atoms are bonded to three metal atoms in a triangular arrangement. The oxygen atoms lie fairly close to the triangular planes; the distances to these planes are 0.184, 0.189, 0.140 and 0.272 Å for the oxygen atoms 1 to 4 respectively.

The deviation from closest packing of the anions is demonstrated by the considerable divergence of the close O–O distances. There are only 8 or 9 oxygen neighbours within 3.0 Å and the twelfth-nearest oxygen atom, completing the coordination sphere, is found at a distance up to 3.33 Å. With three exceptions, all oxygen atoms are more than 2.67 Å apart. The two shortest O–O separations, 2.411 and 2.507 Å, are along those edges in the WO₆ octahedron which are shared with equivalent octahedra. Shortening of such edges is a well-known phenomenon in polyhedra around high-valence cations when no electrons are available for direct metal–metal bonding. The effect should be less pronounced in the copper–oxygen polyhedra because of the probably lower charge on the copper atoms. Although the third shortest O–O distance (2.625 Å) is along one of the shared edges in the CuO₆ octahedron, the other shared edge, O(1)–O(1), is seen from Table 6 to be considerably longer than normal. This shortening and lengthening of the shared edges can be considered as being mainly an effect of the parallel displacement of the oxygen layers around the copper atoms, mentioned above. In FeWO₄ the same shared-edge shortening is observed within the tungsten polyhedra (O–O, 2.436 Å) but is much less pronounced in the FeO₆ octahedra where the O–O distances along

the common edges are 2.918 Å, insignificantly shorter than the average length of the edges in these octahedra, 2.985 Å.

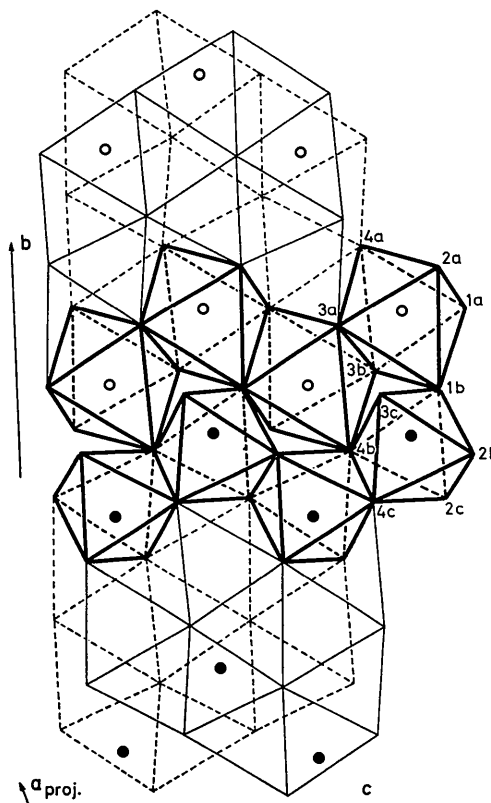


Fig. 1. The structure of CuWO₄ projected on to the *bc* plane. Three pseudo-hexagonal oxygen layers are indicated as triangular nets: in the upper part of the Figure two layers surrounding copper atoms (open circles) and in the lower part two layers enclosing tungsten atoms (filled circles). The top layer (full lines) of the upper part continues as the bottom layer (hatched lines) in the lower half of the Figure. For clarity the unit-cell axes have been displaced from the origin which is situated halfway between the two upper left tungsten atoms. The *a* direction is into the plane of the paper.

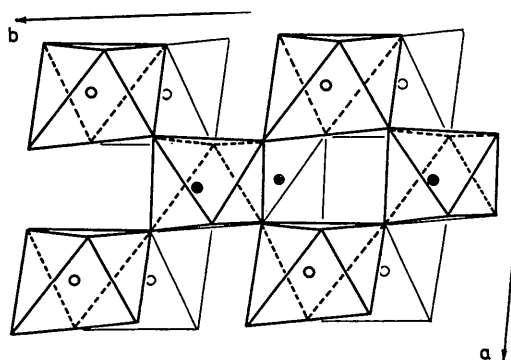


Fig. 2. The structure of CuWO₄ depicted as linked octahedra, viewed along the *c* axis. Open and filled circles represent copper and tungsten atoms, respectively.

The root-mean-square components of thermal vibrations along the ellipsoid axes are given in Table 4. It is seen that the anisotropy is rather small for the tungsten atoms and hardly significant for the oxygen atoms due to the large standard deviations. The values for the copper atom, however, indicate a thermal ellipsoid of nearly rotational symmetry and with one axis considerably longer than the other two. This long ellipsoid axis is directed approximately perpendicular to the plane of Fig. 3. The vibrational amplitude for the copper atom is thus highest perpendicular to the short Cu-O bonds. This is reasonable and was found to be the case also in CuO (Åsbrink & Norrby, 1970).

The authors are grateful for the continued interest shown by Professor A. Magnéli in our studies of copper

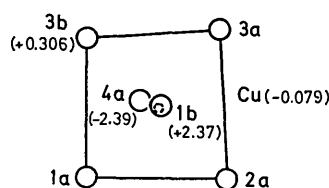


Fig. 3. The coordination around copper projected on to the plane defined by the centres of the oxygen atoms (1a), (2a) and (3a). The values in parentheses are the distances, in Å, of the other atoms above (+) or below (-) this plane.

tungsten oxides. We wish to express appreciation to Mrs E. Hildebrand and Mrs G. Winlöf for collecting the bulk of the diffractometer intensities and for other laboratory assistance. This investigation has been financially supported by the Swedish Natural Science Research Council. Grants for the use of an IBM 360/75 computing system were received from the Computer Division of the Swedish Rationalization Agency (Kungl. Statskontoret). The use of the IBM 1800 machine was made possible by the Tri-Centennial Fund of the Bank of Sweden. One of us (E.G.) expresses appreciation for a grant received from the American-Scandinavian Foundation.

References

- ABRAHAMS, S. C., BERNSTEIN, J. L. & JAMIESON, P. B. (1968). *J. Chem. Phys.* **48**, 2619.
 BUSING, W. R., MARTIN, K. O. & LEVY, H. A. (1962). *ORFLS. A Fortran Crystallographic Least-Squares Program*. Report ORNL-TM-305, Oak Ridge National Laboratory, Oak Ridge, Tennessee.
 BUSING, W. R., MARTIN, K. O. & LEVY, H. A. (1964). *ORFFE*. Report ORNL-TM-306, Oak Ridge National Laboratory, Oak Ridge, Tennessee.
 CHICHAGOV, A. P., ILYUKHIN, V. V. & BELOV, N. V. (1966). *Dokl. Akad. Nauk SSSR*, **166**, 87.
 COPPENS, P., LEISEROWITZ, L. & RABINOVICH, D. (1965). *Acta Cryst.* **18**, 1035.

Table 6. *Interatomic distances and bond angles*

The Table gives metal-oxygen and metal-metal distances (in Å) as well as the bond angles (in degrees) at the metal atoms and the corresponding oxygen-oxygen separation (in Å). The standard deviations in the last digit are in brackets.

		Within CuO ₆ octahedra					
Cu-O(2a)	1.961 (4)	O(1b)-Cu-O(1a)	89.7 (2)	3.135 (9)	O(2a)-Cu-O(4a)	101.2 (2)	3.332 (6)
-O(1a)	1.967 (5)	O(1b)-Cu-O(2a)	87.2 (2)	3.064 (6)	O(3b)-Cu-O(3a)	82.6 (2)	2.625 (9)
-O(3a)	1.978 (5)	O(1b)-Cu-O(3b)	79.5 (2)	2.865 (6)	O(3b)-Cu-O(4a)	92.1 (2)	3.131 (6)
-O(3b)	1.997 (5)	O(1b)-Cu-O(3a)	85.7 (2)	3.033 (7)	O(3a)-Cu-O(4a)	94.5 (2)	3.179 (6)
-O(4a)	2.340 (5)	O(1a)-Cu-O(2a)	90.4 (2)	2.788 (6)	O(1b)-Cu-O(4a)	171.5 (2)	4.777 (6)
-O(1b)	2.450 (5)	O(1a)-Cu-O(3b)	88.5 (2)	2.765 (7)	O(1a)-Cu-O(3a)	170.6 (2)	3.932 (7)
		O(1a)-Cu-O(4a)	88.8 (2)	3.026 (7)	O(2a)-Cu-O(3b)	166.7 (2)	3.932 (6)
		O(2a)-Cu-O(3a)	97.5 (2)	2.963 (6)			
		Within WO ₆ octahedra					
W-O(1b)	1.760 (5)	O(1b)-W-O(2c)	94.3 (2)	2.784 (6)	O(2b)-W-O(4c)	82.4 (2)	2.769 (6)
-O(4b)	1.816 (5)	O(1b)-W-O(2b)	98.4 (2)	2.840 (6)	O(3c)-W-O(4c)	84.7 (2)	2.743 (6)
-O(3c)	1.845 (4)	O(1b)-W-O(3c)	103.2 (2)	2.826 (6)	O(3c)-W-O(4b)	98.0 (2)	2.762 (6)
-O(2b)	1.988 (4)	O(1b)-W-O(4b)	100.9 (2)	2.757 (7)	O(4c)-W-O(4b)	76.4 (2)	2.507 (9)
-O(2c)	2.028 (4)	O(2c)-W-O(2b)	73.8 (2)	2.411 (9)	O(1b)-W-O(4c)	172.0 (2)	3.959 (6)
-O(4c)	2.208 (5)	O(2c)-W-O(4c)	78.2 (2)	2.676 (6)	O(2c)-W-O(3c)	159.5 (2)	3.811 (6)
		O(2c)-W-O(4b)	88.9 (2)	2.697 (6)	O(2b)-W-O(4b)	155.0 (2)	3.714 (6)
		O(2b)-W-O(3c)	93.0 (2)	2.781 (6)			
		Between polyhedra					
Cu-Cu	2.986 (2)	W-W	3.172 (1)	O(1)-W	1.760 (5)	O(3)-W	1.845 (4)
-Cu	3.150 (2)	-W	3.212 (1)	-Cu	1.967 (5)	-Cu	1.978 (5)
				-Cu	2.450 (5)	-Cu	1.997 (5)
		Cu-W, W-Cu	3.437 (1)	O(2)-Cu	1.961 (4)	O(4)-W	1.816 (5)
			3.468 (1)	-W	1.988 (4)	-W	2.208 (5)
			3.507 (1)	-W	2.028 (4)	-Cu	2.340 (5)
			3.524 (1)				
			3.549 (1)				
			3.636 (1)				
			3.826 (1)				
			4.117 (1)				

- CROMER, D. T. (1965). *Acta Cryst.* **18**, 17.
 DACHS, H., STOLL, E. & WEITZEL, H. (1967). *Z. Kristallogr.* **125**, 120.
 GEBERT, E. & KIHLBORG, L. (1967). *Acta Chem. Scand.* **21**, 2575.
 GEBERT, E. & KIHLBORG, L. (1969). *Acta Chem. Scand.* **23**, 221.
 HANSON, H. P., HERMAN, F., LEA, J. D. & SKILLMAN, S. (1964). *Acta Cryst.* **17**, 1040.
 KEELING, R. O. JR (1957). *Acta Cryst.* **10**, 209.
 ÜLKÜ, D. (1967). *Z. Kristallogr.* **124** 192.
 ZACHARIASEN, W. H. (1963). *Acta Cryst.* **16**, 1139.
 ZACHARIASEN, W. H. & PLETTINGER, H. A. (1965). *Acta Cryst.* **18**, 710.
 ÅSBRINK, S. & NORRBY, L.-J. (1970). *Acta Cryst. B* **26**, 8.
 ÅSBRINK, S. & WERNER, P.-E. (1966). *Acta Cryst.* **20**, 407.

Acta Cryst. (1970). B26, 1026

The Structure of 3,4:7,8-Dibenzotricyclo[4.2.0.0^{2,5}]octa-3,7-diene

BY B. L. BARNETT AND RAYMOND E. DAVIS

Department of Chemistry, The University of Texas at Austin, Austin, Texas 78712, U.S.A.

(Received 24 July 1969)

The structure of 3,4:7,8-dibenzotricyclo[4.2.0.0^{2,5}]octa-3,7-diene has been studied by single-crystal diffraction techniques. This compound (C₁₆H₁₂) crystallizes in the monoclinic space group *P*2₁/*c*, with *a* = 10·716, *b* = 7·706, *c* = 6·719 Å, β = 98·37°, *Z* = 2. The structure was determined by the symbolic addition procedure and was refined by block-diagonal least squares methods to a final *R* value of 0·069, including hydrogen atoms. The molecule is in the *anti* configuration, with two planar benzocyclobutene systems on opposite sides of a planar central cyclobutane ring. The bridgehead bond, common to two four-membered rings, is found to be unusually long, 1·593 Å.

Introduction

Since the early fifties, transition metal complexes and complex intermediates have become widely used for producing many organic compounds which would have been difficult or impossible to make otherwise. In producing benzocyclobutadiene and cyclobutadiene (Emerson, Watts & Pettit, 1965) by such methods many new and interesting compounds have been isolated. Three dimers of benzocyclobutadiene have been so prepared (Emerson *et al.*, 1965; Avram, Dinu & Nenitescu, 1959; Cava & Napier, 1957). Formation of these dimeric products of benzocyclobutadiene is dependent upon different transition metals which act as catalysts. The title compound, hereinafter referred to as DBTCO, is the one of these dimers which was prepared using nickel tetracarbonyl as a catalyst (Avram *et al.*, 1959; Cava & Napier, 1957). This compound has so far only been isolated in one configuration, which has been shown by Griffin & Weber (1961) to be *anti*.

In other systems containing four-membered rings, the ring is in either the planar or the puckered conformation, and often exhibits unusually long carbon-carbon bond distances (Adman & Margulis, 1968). Fusion of other rings to four-membered rings seems to have varying effects on the cyclobutane ring geometry, depending on the nature of the fusing rings (Margulis, 1965; Einstein, Hosszu, Longworth, Rahn & Wei, 1967; Adman, Gordon & Jensen, 1968; Camerman, Weinblum & Nyburg, 1969; Camerman & Nyburg, 1969; Barnett & Davis, 1970). The present investigation was undertaken to determine the effect on ring

geometry of fusion of cyclobutane with two highly strained four-membered rings.

Experimental

Suitable crystals for X-ray studies were obtained by recrystallization of a sample of DBTCO, kindly supplied by Professor R. Pettit, from an aqueous acetone solution. These crystals are clear, light gray bulky plates. The crystal used was approximately 0·068 × 0·126 × 0·194 mm with [001] parallel to the φ axis of the goniometer. All X-ray studies were carried out using a General Electric XRD-5 diffractometer equipped with a single crystal orienter, with Cu Kα radiation, and at ambient room temperature (~21 °C).

Plots of the intensity weighted reciprocal lattice were used to determine the crystal symmetry. Cell constants were obtained from least-squares refinement using 27 independent 2θ measurements.

Crystal data

3,4:7,8-Dibenzotricyclo[4.2.0.0^{2,5}]octa-3,7-diene, C₁₆H₁₂. *a* = 10·716 (2), *b* = 7·706 (1), *c* = 6·719 (1) Å, β = 98·37 (2)°. Systematic absences: *h*0*l*, *l* = 2*n* + 1; 0*k*0, *k* = 2*n* + 1. Space group: *P*2₁/*c* (No. 14). *Z* = 2; *F*(000) = 216; *V* = 548·86 Å³; μ = 5·38 (Cu Kα). λ(Cu Kα_{mean}) = 1·54178, Cu Kα₁ = 1·54050, Cu Kα₂ = 1·5433 Å.

Three-dimensional single-crystal intensity data were collected by the stationary-crystal stationary-counter method to the limit 2θ = 130° (Cu Kα₁), using a balanced nickel-cobalt filter pair. Of the 925 in-